

Article

Wear Resistant Nanocomposites Based on Biomedical Grade UHMWPE Paraffin Oil and Carbon Nano-Filler: Preliminary Biocompatibility and Antibacterial Activity Investigation

Michelina Catauro ^{1,*}, Cristina Scolaro ², Giovanni Dal Poggetto ³, Severina Pacifico ⁴ and Annamaria Visco ^{2,5,*}

¹ Department of Engineering, University of Campania “Luigi Vanvitelli”, Via Roma 29, I-81031 Aversa, Italy

² Department of Engineering, University of Messina, C.da Di Dio, 98166 Messina, Italy; cscolaro@unime.it

³ Ecoricerche, Srl, Via Principi Normanni, 81043 Capua (CE), Italy; giogiodp@hotmail.it

⁴ Department of Environmental, Biological and Pharmaceutical Sciences and Technologies; University of Campania “Luigi Vanvitelli”, Via Vivaldi 43, 81100 Caserta, Italy; severina.pacifico@unicampania.it

⁵ Istituto per i Polimeri, Compositi e Biomateriali - CNR IPCB, Via Paolo Gaifami 18, 9-95126 Catania, Italy

* Correspondence: michelina.catauro@unicampania.it (M.C.); avisco@unime.it (A.V.);

Tel.: +39-082/5010360 (M.C.); Tel.: +39-090-676-5249/3808 (A.V.)

Received: 25 March 2020; Accepted: 18 April 2020; Published: 22 April 2020



Abstract: In the present paper, we investigate the effectiveness of nanocomposites (composed of ultra-high molecular weight polyethylene (UHMWPE) mixed with carbon nano-filler (CNF) and medical grade paraffin oil (PO), from the biological point of view. Wear measurements were carried out without (air) and with lubricant (distilled water, natural, and artificial lubricant), and antibacterial activity and cytotoxicity were evaluated. The results highlighted that the presence of CNF is important in the nanocomposite formulation because it reduces the wear rate and prevents oxidative degradation during its processing. An amount of 1.0 wt % of CNF is best because it reaches the optimal distribution within the polymeric matrix, resulting in the best wear resistant, bio-active, and anti-bacterial nanocomposite among all investigated samples.

Keywords: UHMWPE; nanocomposites; wear tests; FTIR; bio-activity

1. Introduction

Biomedical grade ultra high molecular weight polyethylene (UHMWPE) has high wear/abrasion resistance toughness and biocompatibility features and has been the standard material for joint replacement (JR) in artificial knees and hips prosthesis for more than half a century to date [1].

However, UHMWPE is considered the weakest part of the artificial joint: during its use, plastic debris is produced from the wear mechanism. Debris gives rise to adverse reactions, which lead to osteolysis and aseptic loosening of the prosthesis causing the failure of the entire joint, and hence the revision surgery [2]. After implantation in the human body, indeed, UHMWPE undergoes degrading mechanisms in the physiological environment and for the sliding friction against the other metallic or ceramic harder components of the artificial mobile prostheses [3–5]. For this reason, the research of the last years is highly focused on the improvement of mechanical resistance of this material during its use in vitro [6–9]. Highly cross-linked ultra-high-molecular-weight polyethylene was observed to be the most clinically promising in total joint arthroplasty owing to its high wear resistance [10].

Cross-linking of UHMWPE can be initiated by free radicals, which can be introduced by various methods such as exposure to ionizing radiation (i.e., gamma rays, electron beam) or the incorporation

of chemical cross-linking agents such as peroxides or silanes [11–14]. Then, an anti-oxidation treatment is required to avoid material oxidation: thermal quenching, thermal annealing, or blending with antioxidant molecules in an opportune amount [15,16]. Vitamin E has been considered as an important antioxidant for UHMWPE, preventing its oxidative degradation, and increasing its wear resistance and fatigue [17]. Anyway, if on the one hand, vitamin E has been effective as a free radical scavenger, on the other hand, it could reduce the cross-linking efficiency [18]. Recently, Oral et al. 2019 combined the consolidation and cross-linking of UHMWPE in one step, such an opportunity to manufacture highly wear and oxidation-resistant joint implant-bearing surfaces with much improved toughness. The presence of Vitamin E in the UHMWPE formulation is important for its oxidative stabilization [19].

With the aim to enhance the mechanical properties of UHMWPE, many other alternative methods have been used, such as the mixing with reinforcements (particles or fibres) [20]. It is known that graphite acts as a lubricant to increase the wear resistance in friction components. For this reason, UHMWPE reinforced with carbon nanofibers (codified as CFR-UHMWPE, or “Poly II”) has been used in orthopedic implants for total hip or total knee arthroplasty (THA/TKA), in 1970 [21]. However, this composite has been withdrawn because of its decreasing in crack resistance, of the fibers pull off on the surface, and other applicative problems. However, these nanocomposites are reconsidered in these last years, owing to the great innovations in the incorporation methods developed in the new experimentation, and also because of the cytocompatibility of the carbon nanofiller [22]. As an example, Puertolas et al. [23] evaluated the influence of carbon nanotubes (CNTs) and graphene as reinforcement fillers in UHMWPE matrix, while Yousef et al. [4] highlighted that carbon nanofiller can lead to an improvement in the wear behavior of biomedical grade polyethylene.

In our previous papers [4,24], we have shown the wear resistance features of nanocomposites made by UHMWPE mixed with 1.0 wt % of carbon nano-filler (CNF) and 2.0 wt % of medical grade paraffin oil (PO). In particular, the carbon nanofiller acts as lubricant while the paraffin oil reduces the typically high viscosity of UHMWPE, favoring its process ability and the filler dispersion [4]. These nanocomposites are innovative because they exhibited an appreciable wear resistance compared with that of graphene-filled nanocomposite, with the advantage of low cost [24].

In the present paper, we investigate the effectiveness of these nanocomposites (which had a wear behavior enhancement) from the biological point of view (antibacterial activity and cytotoxicity). Biocompatibility or cytotoxicity aspects need to be evaluated to determine if adverse reactions could have occurred as the body’s response.

In the nanocomposites, UHMWPE is mixed with PO and CNF. UHMWPE is commonly classified as bio-inert material. PO is a mineral oil and is compatible with the human body; for this reason, it is used in medical applications as well as in cosmetics [25]. Carbon filler instead needs special attention because its particles could migrate, being dangerous for the human body. In particular, if carbon filler contains dangling carbon bonds on their surface, they are considered highly reactive. Similarly, if carbon filler contains traces of residual catalysts, these could be dangerous and reactive as well. Consequently, the high wear resistance of an UHMWPE based nano-composites containing carbon fillers is the key-factor to limit the migration of the filler. In fact, the more resistant to wear the material, the lower the carbon filler load that could migrate outside in the surrounding areas of the nanocomposite. An improvement of wear resistance is very important to minimize the danger of adverse effects on implants. For these reasons, and because to date there are no studies on this aspect of these nanocomposites related to their biological features, we present the wear and biological (bioactivity and anti-bacterial) features of these materials.

2. Materials and Methods

The control sample (code UP) was made of pure medical grade UHMWPE (code GUR1020 Ticona[®], Sulzbach, Germany, molecular weight = $2\text{--}4 \times 10^6$ g/mol and density = 0.93 g/cm³) mixed with 2.0 weight% of pharmaceutical grade Paraffin oil (Sella Pharmaceutical and Chemical Laboratory, Schio (Vi), Italy), added to UHMWPE to favor its mixing and processing.

Nanocomposites were obtained by mixing pure UHMWPE (code: U) with paraffin oil (2.0 weight%—code: P) and with carbon nanofiller (0.1–0.5–1.0–1.5–2.0 weight%—code: C) in a ball milling for 30 min at 20 Hz. The nanocomposites were identified with the code UPC with a number indicating the weight% of carbon nano-filler (code CNF). The mixing with ball mill was useful in order to induce a high degree of the filler dispersion inside the polymeric matrix, which improves both the mechanical and thermal features of the nanocomposites [5].

CNF powder was obtained by milling short carbon fibers supplied by Zoltek (Bridgeton, MO, USA) in a ball milling (mod. Retsch-MM301, 30 cycles of 10 min at 50 rpm, Retsch, Haan, Germany).

Polymeric sheets were prepared by compression molding (mod. PM 20/200, Campana S.R.L, Veduggio (MB), Italy) in a laboratory press at 200 °C for 20 min, at pressure of 20 MPa. The geometry was 60 mm × 60 mm, 2 mm thick.

More details about the material's preparations have already been explained in detail in our previously published paper [5].

2.1. Materials Characterization

A pin-on-disc wear tester was used to perform wear resistance measurements in different lubricants: no lubricant (air), distilled water, artificial (simulated synovial fluid), and natural lubricant (bovine serum) [24]. Both artificial and natural lubricants were chosen in order to approximate the biological conditions of a human joint. The polymeric samples employed in this test had square shapes: 20 mm × 20 mm, and 2 mm thick. The pin was a ruby corundum grinding stones (code: M.2145 and 3 mm in diameter), at room temperature. The use of such hard material allowed us to obtain the “accelerate wear test”, in which we observed a noteworthy weight loss and improvement in the accuracy of the final result. The pin-on-disc system gives a circular shape wear trajectory with a testing load of 30 N, spin rate of 60 rpm, and test time of 120 min. Anyway, more details are given in a previous paper [4]. For each sample, the specific wear rate W_s (mm³/Nm) was calculated [26]:

$$W_s = \frac{\Delta m}{\rho \times F_n \times L} \quad (1)$$

where Δm (mg) is the mass loss of the specimen, ρ (g/mL) is the density (see values listed in Table 1), F_n (N) is the normal load, and L (m) is the total sliding distance. The mass loss was evaluated by a high sensitivity electronic weighing balance (mod. Explorer pro EP 214C, ASTM D1505 International standard, OHAUS Corporation, Parsippany, NJ, USA, accuracy: 10^{−4} g).

Three tracks of different diameter were evaluated after the two hours in each sample and the W_s value was the average of these three tracks. Besides, the wear test was performed in four different media: no lubricant (air), distilled water, artificial lubricant (or simulated synovial fluid), and natural lubricant (or bovine serum). The final value of the specific wear rate W_s was determined with the average of the W_s values of n.9 polymeric sheets (for each nanocomposite obtained) in the four different media.

Data presentation and statistical analysis: the mean differences and standard deviations of specific wear rate of all the samples, with the CNF filler amount, in different media (air, distilled water, and artificial and natural lubricant), were calculated. The media (air, distilled water, and artificial and natural lubricant) and the CNF (%) nanocomposites were the independent variables. The data were first verified with the D'Agostino & Pearson test for the normality of the distribution and the Levene test for the homogeneity of variances. The data were normally distributed and homogenous; therefore, they were statistically analyzed using two-way analysis of variance (ANOVA) and Bonferroni post hoc test for multiple comparisons at a level of significance set at $p < 0.05$ (Prism 8.4.1; GraphPad Software, Inc., La Jolla, CA, USA).

Artificial lubricant was a simulated synovial fluid (SSF), formed by dissolving 0.3 wt % of hyaluronic acid in a phosphate buffered saline solution at pH = 7.4. The electrolyte concentration was as follows: Na⁺ 153.1 mM, K⁺ 4.2 mM, Cl[−] 139.6 mM, phosphate buffer 9.6 mM [24].

Bovine serum, or natural lubricant, is a synovial fluid extracted from the joint of a young bovine according to current legislation. It was kept in a refrigerator at $-20\text{ }^{\circ}\text{C}$ before each test.

Density was calculated by means of the precision balance before described, equipped like a hydrostatic system that follows the Archimede's principle. Each test was performed at room temperature for five minutes. The density (ρ) was evaluated from dry (P_{dry}) and wet (P_{wet}) weight of the sample, before and after the immersion in ethanol, with the following equation:

$$\rho = \frac{P_{dry}}{P_{dry} - P_{wet}} \rho_{eth} \quad (2)$$

where ρ_{eth} is equal to 0.790 g/cm^3 . The resulting density value of each sample was the average of $n = 3$ measurements.

2.2. Fourier Transform Infrared (ATR/FTIR) Spectroscopy

The chemical composition of the different materials was analyzed by attenuated total reflectance Fourier transform infrared (ATR/FTIR) spectroscopy (Shimazu, Tokyo, Japan). The spectra were obtained with the Prestige-21 FTIR spectrometer equipped with an AIM-8800 infrared microscope (Shimadzu, Tokyo, Japan), using the incorporated 3mm diameter Ge attenuated total reflectance (ATR) semicircular prism. Furthermore, the spectra were recorded with an incident angle of 30° with a resolution of 4 cm^{-1} (64 scan) and were in the range of $650\text{--}4000\text{ cm}^{-1}$. The Prestige software (IRsolution, version 1.10, Shimadzu, Tokyo, Japan) was used to further analyze the spectra.

Table 1. Values of reference UP and of the nanocomposites at different immersion times. U, ultra high molecular weight polyethylene (UHMWPE); P, paraffin oil; C, carbon nanofiller.

Sample Code	Density (g/mL)
UP	0.866 ± 0.001
UPC 0.1 wt %	0.864 ± 0.001
UPC 0.5 wt %	0.862 ± 0.002
UPC 1.0 wt %	0.862 ± 0.004
UPC 1.5 wt %	0.863 ± 0.002
UPC 2.0 wt %	0.865 ± 0.002

2.3. Fourier Transform Infrared (ATR/FTIR) Spectroscopy

In order to evaluate the bioactivity, the materials were soaked in simulated body fluid (SBF) with ion concentration nearly equal to those in human blood plasma [27].

Table 2. Body fluid (simulated body fluid, SBF) composition.

Ion	Concentration/mol m ³	
	SBF	Human Blood Plasma
Na ⁺	142.0	142.0
K ⁺	5.0	5.0
Mg ²⁺	1.5	1.5
Ca ²⁺	2.5	2.5
Cl ⁻	147.8	103.0
HCO ₃ ⁻	4.2	27.0
HPO ₄ ²⁻	1.0	1.0
SO ₄ ²⁻	0.5	0.5

The SBF was prepared from NaCl, NaHCO₃, KCl, MgCl₂, 1 M HCl, CaCl₂·6H₂O, and Na₂SO₄ (Sigma-Aldrich, St. Louis, MO, USA) with a concentration that was suggested by Kokubo [27] (Table 2). The pH of the buffer was adjusted to pH 7.4 using 1 M HCl. The solution was exchanged every two days

to avoid depletion of the ionic species in the SBF owing to the formation of biominerals. After 21 days of exposure at 37 °C, the samples were removed from the SBF and air-dried in a desiccator.

Fourier transform infrared (ATR/FTIR) spectroscopy (Shimadzu, Tokyo, Japan) was used to observe the characteristic peaks of hydroxyapatite layer on the surface of materials.

2.4. Antibacterial Activity

Escherichia coli, Gram-negative (ATCC 25922) and *Enterococcus faecalis*, Gram-positive (ATCC 29212) were used to evaluate the antibacterial properties of UHMWPE with 2.0 wt % of paraffin oil in function of different amounts of carbon nano-filler.

The bacterial culture was diluted in distilled water to produce a bacterial cell suspension of 10×10^5 CFU/mL (where CFU is for colony forming unit). *E. coli* and *E. faecalis* were inoculated in TBX medium (Tryptone Bile X-Gluc) (Liofilchem, Italy) and in Slanetz Bartley agar base (Liofilchem, Italy), respectively.

The materials were incubated against *E. coli* for 24 h at 36 °C and against *E. faecalis* for 24 h at 44 °C. The microbial growth was evaluated by observing the diameter of the inhibition halo (ID). The obtained values are the mean standard (SD) deviation of the measurements carried out on samples analyzed three times.

2.5. Cytotoxicity

3-(4,5-dimethylthiazol-2-yl)-2,5-diphenyl tetrazolium bromide (MTT) assay was used to determine the metabolic activity on NIH-3T3 murine fibroblast cells. For this purpose, cells were grown in Dulbecco's modified Eagle medium supplemented with 10% fetal bovine serum, 50.0 U/mL penicillin, and 100.0 µg/mL streptomycin, at 37 °C in a humidified atmosphere containing 5% CO₂. When cells were seeded at a density equal to 5.0×10^5 per well, onto six-well plates, they are directly exposed to disks of synthesized material. After 6, 12, and 24 h of incubation, cells were treated with MTT (500 µL; 0.50 mg/mL), previously dissolved in culture media, for 2 h at 37 °C in a 5% CO₂ humidified atmosphere. The MTT solution was then removed and dimethyl sulphoxide (DMSO) was added to dissolve the original formazan. Finally, the absorbance at 570 nm of each well was determined using a Victor3 Perkin Elmer fluorescence and absorbance reader. The cell viability was expressed as a percentage of mitochondrial redox activity of the cells directly exposed to powders, compared with an unexposed control.

3. Results and Discussion

3.1. Wear Behavior

Considering a comparison of the specific wear rate (*Ws*) of the control sample (UP) and the nanocomposites in different lubricating media reported in Figure 1, we have the following trend of *Ws*:

$$\text{Air} > \text{Water} > \text{Artificial Lubricant} > \text{Natural Lubricant}$$

Thus, the natural bovine serum has the best lubricant effect, while the air (or no lubricant) has the worst, as expected. The highest reduction occurs in natural lubricant, for all the samples, regardless of the CNF amount. In particular, the decrease was −78.03% in the UP control sample (from 47.3×10^{-6} mm³/Nm to 10.39×10^{-6} mm³/Nm, $p < 0.0001$) and −89.01% (from 40.78×10^{-6} mm³/Nm to 4.48×10^{-6} mm³/Nm, $p < 0.0001$) in the UPC 0.5 wt %, and −93.29% (from 37.91×10^{-6} mm³/Nm to 2.54×10^{-6} mm³/Nm, $p < 0.0001$) in the UPC 1.0 wt %. These data suggest that UPC 1.0 wt % nanocomposite exhibits the highest wear rate reduction, regardless of the lubricating media. In fact, for higher filler amounts, the wear rate improves again.

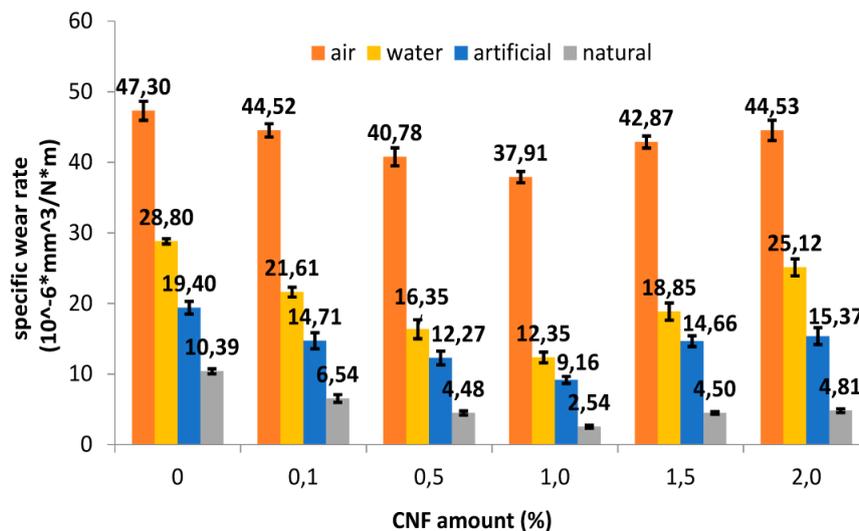


Figure 1. Wear rate of all of the samples vs. the carbon nano-filler (CNF) amount in different media (air, distilled water, and artificial and natural lubricant). The differences between the groups are statistically significant according to two-way analysis of variance (ANOVA) interaction and post hoc Bonferroni test, $p < 0.0001$.

This is because the filler dispersion changes with the different loads; at a low amount, such as 0.1 or 0.5 wt %, the dispersion is poor, while at higher amounts, such as 1.5 or 2.0 wt %, the great filler amount let to a filler agglomeration, which forms weak points that act as localized stress concentration points with an increases of specific wear rate improvement for an improvement of weight loss [28]. The inferential analysis revealed statistically significant differences in the effect of the media (air, distilled water, and artificial and natural lubricant) on the nanocomposites ($p < 0.0001$). The statistical analysis carried out also showed that both variables (carbon nanofiller content and type of medium used) have a statistically significant effect on the wear data obtained (interaction, $p < 0.0001$).

3.2. Fourier Transform Infrared (ATR/FTIR) Spectroscopy

The chemical interactions among the components in the materials and its chemical structure were evaluated using ATR/FTIR spectroscopy. In Figure 2 are reported the spectra of all samples with different percentages of carbon nano-filler (curves b, c, d, e) compared with UHMWPE with paraffin oil and without carbon nano-filler (curve a).

In all spectra, the bands at 2930 cm^{-1} and 2847 cm^{-1} are observed owing to the $-\text{CH}-\text{CH}$ asymmetric and symmetric stretching modes [29,30]. In addition, the peak at 1465 cm^{-1} was attributed to $-\text{CH}$ asymmetrical bending, while the $\text{C}-\text{C}$ stretching vibrations could be assigned at 1242 cm^{-1} [29,30]. Furthermore, in the spectrum of UHMWPE with paraffin oil and without carbon nano-filler (curve a), the peaks at 1743 cm^{-1} and 721 cm^{-1} were detectable. The peak at 721 cm^{-1} was reported to be related to the asymmetric angular deformation of CH_2 groups in mineral oil [31]. The band at 1743 cm^{-1} could be attributable to an oxidative degradation process that occurs during the high temperature processing. In fact, the peak at 1743 cm^{-1} is attributed to the absorption of carbonyl species ($\text{C}=\text{O}$) [32], while oxygen bearing functionalities such as $\text{C}-\text{O}$ stretching could be assigned at 1242 cm^{-1} . Observing the spectra of the polymer containing the different percentage of carbon nano-filler, the peak at 1743 cm^{-1} disappeared, and it was weakly detectable in UPC 2.0 wt % CNF spectrum (Figure 2 panel E). This could be because of the ability of carbon nano-filler (CNF) to improve resistance to oxidative degradation. Indeed, the $\text{C}-\text{O}$ stretching vibration was also more pronounced in UPC 2.0 wt % CNF spectrum. The different change in shape and absorbance of the peak at 1242 cm^{-1} could be the result of CNF $\text{C}-\text{C}$ stretching modes overlapping [33].

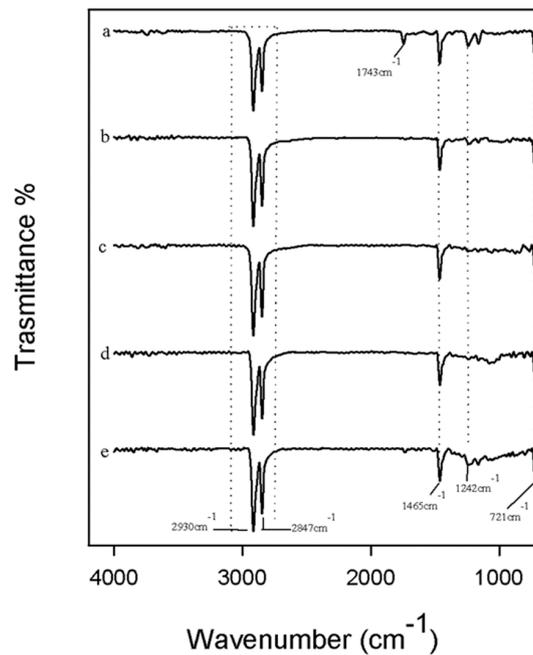


Figure 2. Total reflectance Fourier transform infrared (ATR/FTIR) spectroscopy spectra of (a) UP; (b) UPC 0.1 wt % CNF; (c) UPC 0.5 wt % CNF; (d) UPC 1.0 wt % CNF; and (e) UPC 2.0 wt % CNF. U, ultra high molecular weight polyethylene (UHMWPE); P, paraffin oil; C, carbon nanofiller.

3.3. Bioactivity Test

The bioactivity properties of the different materials were evaluated using Kokubo's test. The materials were soaked in simulated body fluid (SBF) for 21 days, and after this exposure time, the formation of the hydroxyapatite on the surfaces of all samples was detected by ATR/FTIR.

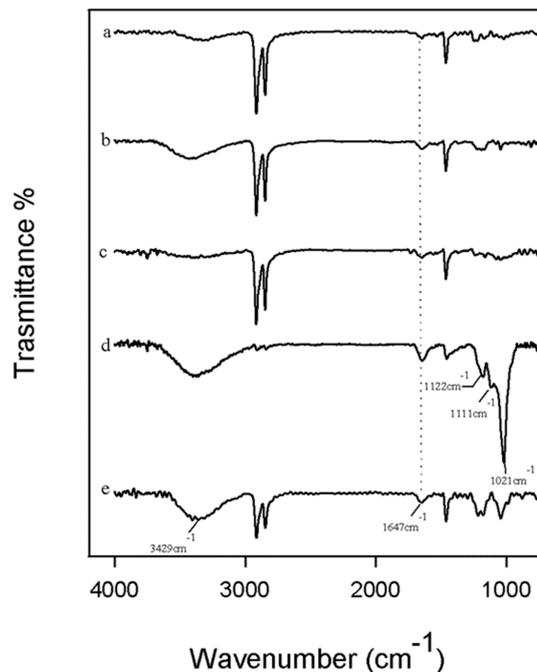


Figure 3. Total reflectance Fourier transform infrared (ATR/FTIR) spectroscopy spectra of (a) UP; (b) UPC 0.1 wt % CNF; (c) UPC 0.5 wt % CNF; (d) UPC 1.0 wt % CNF; and (e) UPC 2.0 wt % CNF after 21 days in simulated body fluid (SBF).

In Figure 3, the spectra of all materials are reported, in which it is possible to observe the peaks at 3429 and 1647 cm^{-1} that correspond to the $-\text{OH}$ groups [34,35]. Comparing all materials, in the spectra of UHMWPE with paraffin oil and with $1.0\text{ wt } \%$ of carbon nano-filler (curve d), the typical bands of the hydroxyapatite layer are clearly visible. In fact, the peak at 1021 cm^{-1} is the result of the absorption of the stretching mode of the PO_4^{3-} ions [36–38]. In addition, the P–O bond of the phosphate group appears at 1122 and 1111 cm^{-1} [39]. Furthermore, only in this spectrum, the typical peaks of the polymer are not very intense, which is probably because the material's surface is covered by a hydroxyapatite layer that is very thick compared with the others [40,41]. In conclusion, all materials seem to be bioactive because the hydroxyapatite bands are present in all spectra, but when carbon nano-filler is added to $1.0\text{ wt } \%$, the hydroxyapatite peaks are more intense, suggesting that this material is the most bioactive of all.

According to the wear resistance test result, the optimal wear performance and optimal bioactivity are noticed in the same nanocomposite, containing $1.0\text{ wt } \%$ of carbon nano-filler. This result confirms that the filler dispersion is a key parameter to achieve the best mechanical as well as biological response, as discussed in Section 3.1.

3.4. Antibacterial Activity

The materials with the same percentage of paraffin oil and several amounts of carbon nano-filler were incubated with gram-positive and gram-negative bacterial strains. In Figure 4A, the representative images of both bacteria are reported, while the bacterial growth is shown in Figure 4B. The UP material with only $2.0\text{ wt } \%$ of paraffin oil was used as control in order to evaluate the effects of carbon nano-filler on bacterial growth. In all plates, after 24 h of incubation against *E. coli* and *E. faecalis*, the inhibition halo (ID) was not observed, regardless of the carbon nano-filler content. Therefore, the results show that the materials do not have toxic effects against gram-positive and gram-negative bacterial strains. In addition, the materials during the incubation period did not release any product that could damage the bacteria growth, suggesting that they could be used as optimal materials for joint replacement (JR) in artificial knees and hips prosthesis.

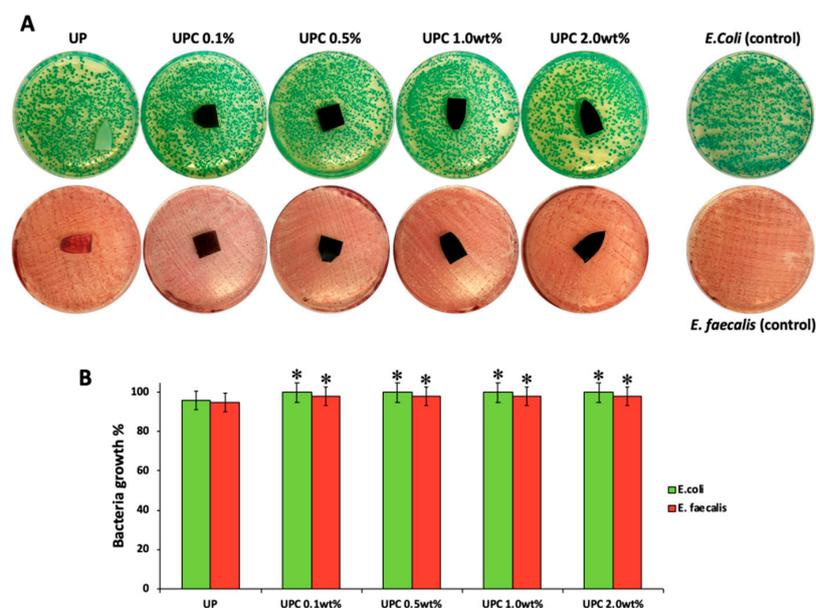


Figure 4. (A) Representative image of *E. coli* and *E. faecalis* incubated with all materials. (B) Bacteria growth (%) of *E. coli* and *E. faecalis* incubated with UP, UPC $0.1\text{ wt } \%$, UPC $0.5\text{ wt } \%$, UPC $1.0\text{ wt } \%$, and UPC $2.0\text{ wt } \%$. Values are the mean SD of measurements carried out on samples analyzed three times. The means and S.D. are shown. * $p < 0.05$ vs. the bacteria control treated without materials.

3.5. Cytotoxicity

The nanocomposites were investigated for their biocompatibility through MTT-direct contact test. The data (Figure 5) acquired highlighted that all the samples avoided cytotoxic effects, and that the increase in amount percentage in carbon nano-filler positively affected NIH-3T3 viability. In fact, all the samples appeared to induce a cell growth increase with respect to UP material, reaching its maximum by the UPC1% sample. This latter showed a percentage increase in cell viability equal to 10.2% compared with UP. Thus, the incorporation of CNFs in a small dose level percentage seemed to not exert evident redox mitochondrial modification. Previous studies evidenced that multiwall carbon nanotubes caused a significant time- and dose-dependent decrease in cells' viability [42,43], and that they could be toxic at sufficiently high concentrations. Indeed, it was reported that carbon nanomaterials present significantly different cytotoxicity depending on their physicochemical properties, including size, length, shape, and surface area [44–46]. The lack of filler coalescence, defined as a consequence of CNF incorporation, suggested the constitution of an improved UHMWPE-based material, in which the absence of debris could favour the slowing down of inflammatory and oxidative stress effects commonly ascribed to these materials [47–52].

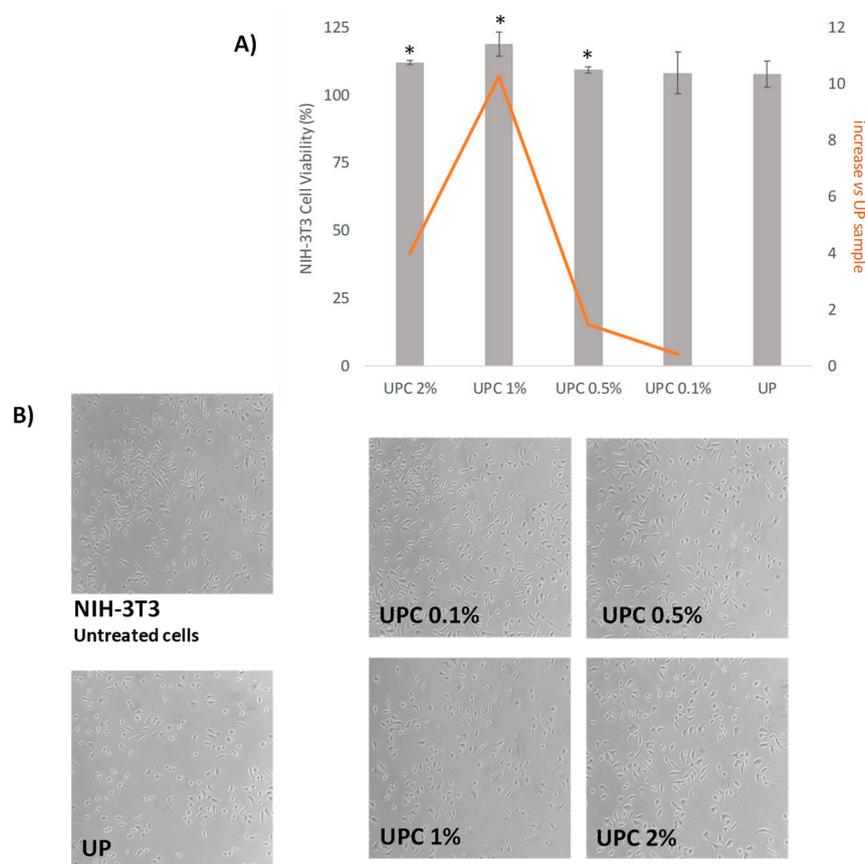


Figure 5. (A) Data from MTT assay expressed as NIH-3T3 cells viability (%) vs. untreated cells. Values are the mean \pm SD of two independent experiments performed in triplicate. * $p < 0.05$ vs. untreated cells. The increase in cell viability vs. UP sample is also shown. (B) Representative cell morphology images of NIH-3T3 cells after treatment with investigated samples. Images were acquired by Inverted Phase Contrast Brightfield Zeiss Primo Vert Microscope.

4. Conclusions

In this paper, we investigated the biological and antibacterial activity of wear resistant nanocomposites based on biomedical grade UHMWPE paraffin oil and carbon nano-filler. Paraffin oil favors the workability of the typical high viscosity UHMWPE and the filler dispersion within the

polymer. Carbon nanofiller act as lubricant to lower the friction between the UHMWPE and other counterparts. Furthermore, carbon nano-filler prevents the oxidative degradation of the nanocomposite during its processing, as observed by the FTIR analysis.

The wear test was performed in different lubricating media (no lubricant or air, distilled water, simulated synovial fluid, and bovine serum). The lubricating effect was in the following order: natural lubricant > artificial lubricant > water > air. The more wear resistant nanocomposite resulted it being reinforced with 1.0 wt % of CNF regardless of the lubricating media because of the best filler dispersion within the polymeric matrix. A filler load lower than 1.0 weight percentage, as well as a higher one, are too little and too much, respectively, in order to achieve a homogeneous dispersion and to avoid filler coalescence, respectively. This aspect also reflects the bio-activity response of the materials, which has the best hydroxyapatite production in just the 1.0 wt % sample.

This suggests that it could be applicable in artificial prosthesis. In order to verify in more depth the suitability of these nanocomposites, further studies are underway on the release of components over time into biological fluids.

Author Contributions: For research articles with several authors, a short paragraph specifying their individual contributions must be provided. The following statements should be used “conceptualization, M.C. and A.V.; methodology, M.C., S.P.; software, C.S.; validation, M.C., A.V. and S.P.; formal analysis, C.S., and G.D.P.; investigation, C.S.; resources, A.V.; data curation; M.C.; writing—original draft preparation, A.V.; writing—review and editing, M.C.; visualization, S.P.; supervision, M.C., A.V. All authors have read and agreed to the published version of the manuscript.

Funding: The work was financially supported by V:ALERE 2019 grant support from University of Campania “Luigi Vanvitelli” of CHIMERA.

Acknowledgments: Authors thank dott. Eugenio Pedullà of General surgery and medical-surgical specialties Department A.O.U. Policlinico Vittorio Emanuele-Catania, Italy for his help in conducting statistical analyses.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Gomez-Barrena, E.; Puertolas, J.-A.; Munuera, L.; Konttinen, Y.T. Update on UHMWPE research from the bench to the bedside. *Acta Orthop* **2008**, *79*, 832–840. [[CrossRef](#)] [[PubMed](#)]
2. Zhou, J.; Chakravartula, A.; Pruitt, L.; Komvopoulos, K. Tribological and Nanomechanical Properties of Unmodified and Crosslinked Ultra-High Molecular Weight Polyethylene for Total Joint Replacements. *J. Tribol.* **2004**, *126*, 386. [[CrossRef](#)]
3. Ge, S.; Kang, X.; Zhao, Y. One-year biodegradation study of UHMWPE as artificial joint materials: Variation of chemical structure and effect on friction and wear behavior. *Wear* **2011**, *271*, 2354–2363. [[CrossRef](#)]
4. Yousef, S.; Visco, A.M.; Galtieri, G.; Nocita, D.; Espro, C. Wear behaviour of UHMWPE reinforced by carbon nanofiller and paraffin oil for joint replacement. *Mater. Sci. Eng. C* **2017**, *73*, 234–244. [[CrossRef](#)] [[PubMed](#)]
5. Visco, A.; Yousef, S.; Galtieri, G.; Nocita, D.; Njuguna, J. Thermal, Mechanical and Rheological Behaviors of Nanocomposites Based on UHMWPE/Paraffin Oil/Carbon Nanofiller Obtained by Using Different Dispersion Techniques. *JOM-Springer* **2016**, *68*, 1078–1089. [[CrossRef](#)]
6. Atwood, S.A.; van Citters, D.W.; Patten, E.W.; Furmanski, J.; Ries, M.D.; Pruitt, L.A. Tradeoffs amongst fatigue, wear, and oxidation resistance of cross-linked ultra-high molecular weight polyethylene. *J. Mech. Behav. Biomed. Mater.* **2011**, *4*, 1033–1045. [[CrossRef](#)] [[PubMed](#)]
7. Lewis, G.; Fencl, R.M.; Carroll, M.; Collins, T. The relative influence of five variables on the in vitro wear rate of uncrosslinked UHMWPE acetabular cup liners. *Biomaterials* **2003**, *24*, 1925–1935. [[CrossRef](#)]
8. Bruck, A.L.; KanagaKaruppiyah, K.S.; Sundararajan, S.; Wang, J.; Lin, Z. Friction and wear behavior of ultrahigh molecular weight polyethylene as a function of crystallinity in the presence of the phospholipid dipalmitoyl phosphatidylcholine. *J. Biomed. Mater. Res. Part. B Appl. Biomater.* **2010**, *93B*, 351–358. [[CrossRef](#)]
9. Visco, A.M.; Campo, N. Properties of nanocomposites based on polyethylene (UHMWPE) and carbon nanotubes mixed by high energy ball milling and UV-source irradiated. *Int. J. Polym. Anal. Charact.* **2012**, *17*, 144–157.

10. Muratoglu, O.K.; Bragdon, C.R.; O'Connor, D.O.; Jasty, M.; Harris, W.H.; Gul, R.; McGarry, F. Unified wear model for highly crosslinked ultra-high molecular weight polyethylenes (UHMWPE). *Biomaterials* **1999**, *20*, 1463–1470. [[CrossRef](#)]
11. Visco, A.M.; Campo, N.; Torrisi, L.; Cristani, M.; Trombetta, D.; Saija, A. Electron beam irradiated UHMWPE: Degrading action of air and hyaluronic acid. *Bio-Med. Mater. Eng.* **2008**, *18*, 137–148. [[CrossRef](#)]
12. Gul, R.M. The effects of peroxide content on the wear behavior, microstructure and mechanical properties of peroxide cross-linked UHMWPE used in total hip replacement. *J. Mater. Sci. Mater. Med.* **2008**, *19*, 2427–2435. [[CrossRef](#)] [[PubMed](#)]
13. Atkinson, J.R.; Cicek, R.Z. Silanecrosslinked polyethylene for prosthetic applications Part I. Certain physical and mechanical properties related to the nature of the material. *Biomaterials* **1983**, *4*, 267–275. [[CrossRef](#)]
14. Atkinson, J.R.; Cicek, R.Z. Silanecrosslinked polyethylene for prosthetic applications: II. Creep and wear behaviour and a preliminary moulding test. *Biomaterials* **1984**, *5*, 326–335. [[CrossRef](#)]
15. Oral, E.; GodleskiBeckos, C.; Malhi, A.S.; Muratoglu, O.K. The effects of high dose irradiation on the cross-linking of vitamin E-blended ultrahigh molecular weight polyethylene. *Biomaterials* **2008**, *29*, 3557–3560. [[CrossRef](#)]
16. Visco, A.M.; Campo, N.; Brancato, V.; Trimarchi, M. Influence of thea-tocopherol load and the annealing treatment on the wear resistance of biomedical UHMWPE radiated with electron beam. *Int. J. Polym. Anal. Charact.* **2013**, *18*, 545–556. [[CrossRef](#)]
17. Oral, E.; Christensen, S.D.; Malhi, A.S.; Wannomae, K.K.; Muratoglu, O.K. Wear Resistance and Mechanical Properties of Highly Cross-linked, Ultrahigh-Molecular Weight Polyethylene Doped With Vitamin E. *J. Arthroplast.* **2006**, *21*, 580–591. [[CrossRef](#)]
18. Oral, E.; GodleskiBeckos, C.; Lozynsky, A.J.; Malhi, A.S.; Muratoglu, O.K. Improved resistance to wear and fatigue fracture in high pressure crystallized vitamin E-containing ultra-high molecular weight polyethylene. *Biomaterials* **2009**, *30*, 1870–1880. [[CrossRef](#)]
19. Oral, E.; Doshi, B.N.; Fung, K.; O'Brien, C.; Wannomae, K.K.; Muratoglu, O.K. Chemically Cross-Linked UHMWPE With Superior Toughness. *J. Orthop. Res.* **2019**, *37*, 2182–2188. [[CrossRef](#)]
20. Hussain, M.; Naqvi, R.A.; Abbas, N.; Khan, S.M.; Nawaz, S.; Hussain, A.; Zahra, N.; Khalid, M.W. Ultra-High-Molecular-Weight-Polyethylene (UHMWPE) as a Promising Polymer Material for Biomedical Applications: A Concise Review. *Polymers* **2020**, *12*, 323. [[CrossRef](#)]
21. Panin, S.V.; Kornienko, L.A.; Nguyen, D.A.; Alexenko, V.O.; Ivanova, L.R. Enhancement of mechanical and tribotechnical properties of polymer composites with thermoplastic UHMWPE and PEEK matrices by loading carbon nanofibers/nanotubes. In Proceedings of the 11th International Conference on Mechanics, Resource and Diagnostics of Materials and Structures, AIP Conference Proceedings Mechanics, Resource and Diagnostics of Materials and Structures (Mrdms-2017), Ekaterinburg, Russia, 11–15 December 2017; Eduard, S., Gorkunov, V.E., Sunder, R., Eds.; American Institute of Physics: College Park, MD, USA, 2017; Volume 1915, p. 030015.
22. Kurtz, S.M. *The UHMWPE Handbook*, 1st ed.; Elsevier Academic Press: San Diego, CA, USA; London, UK, 2004.
23. Puértolas, J.A.; Kurtz, S.M. Evaluation of carbon nanotubes and graphene as reinforcements for UHMWPE based composites in arthroplastic applications: A review. *J. Mech. Behav. Biomed. Mater.* **2014**, *39*, 129–145. [[CrossRef](#)] [[PubMed](#)]
24. Visco, A.; Yousef, S.; Scolaro, C.; Espro, C.; Cristani, M. Tribological Behavior of Nanocomposites Based on UHMWPE Aged in Simulated Synovial Fluid. *Polymers* **2018**, *10*, 1291. [[CrossRef](#)] [[PubMed](#)]
25. Fiorito, S.; Serafino, A.; Andreola, F.; Togna, A.; Togna, G. Toxicity and biocompatibilità of carbon nanoparticles. *J. Nanosci. Nanotech.* **2006**, *6*, 1–9. [[CrossRef](#)] [[PubMed](#)]
26. Agarwal, G.; Patnaik, A.; Sharma, R.K. Parametric optimization and three-body abrasive wear behavior of sic chopped glass fiber reinforced epoxy composites. *Int. J. Compos. Mater.* **2013**, *3*, 32–38.
27. Kokubo, T.; Takadama, H. How useful is SBF in predicting in vivo bone bioactivity? *Biomaterials* **2006**, *27*, 2907–2915. [[CrossRef](#)]
28. Harsha, A.P.; Joyce, Y.J. Challenges associated with using bovine serum in wear testing orthopaedic biopolymers. *J. Eng. Med.* **2011**, *225*, 948–958. [[CrossRef](#)]
29. Ahmadi, M.; Masoomi, M.; Safi, S.; Zabihi, O. Interfacial evaluation of epoxy/carbon nanofiber nanocomposite reinforced with glycidyl methacrylate treated UHMWPE fiber. *J. Appl. Polym. Sci.* **2016**, *133*. [[CrossRef](#)]

30. Barick, A.K.; Tripathy, D.K. Effect of nanofiber on material properties of vapor-grown carbon nanofiber reinforced thermoplastic polyurethane (TPU/CNF) nanocomposites prepared by melt compounding. *Compos. Part. A Appl. Sci. Manuf.* **2010**, *41*, 1471–1482. [[CrossRef](#)]
31. Houmard, M.; Nunes, E.H.M.; Vasconcelos, D.C.L.; Berthomé, G.; Joud, J.-C.; Langlet, M.; Vasconcelos, W.L. Correlation between sol-gel reactivity and wettability of silica films deposited on stainless steel. *Appl. Surf. Sci.* **2014**, *289*, 218–223. [[CrossRef](#)]
32. Azam, A.M.; Ali, A.; Khan, H.; Yasin, T.; Mehmood, M.S. Analysis of degradation in UHMWPE a comparative study among the various commercial and laboratory grades UHMWPE. In *IOP Conference Series: Materials Science and Engineering*; IOP Publishing: Bristol, UK, 2016.
33. Ma, H.; Zeng, J.; Realff, M.L.; Kumar, S.; Schiraldi, D.A. Processing, structure, and properties of fibers from polyester/carbon nanofiber composites. *Compos. Sci. Technol.* **2003**, *63*, 1617–1628. [[CrossRef](#)]
34. Ohtsuki, C.; Kokubo, T.; Yamamuro, T. Mechanism of apatite formation on CaOSiO₂P₂O₅ glasses in a simulated body fluid. *J. Non-Cryst. Solids* **1992**, *143*, 84–92. [[CrossRef](#)]
35. Catauro, M.; Barrino, F.; Poggetto, G.D.; Crescente, G.; Piccolella, S.; Pacifico, S. New SiO₂/Caffeic acid hybrid materials: Synthesis, spectroscopic characterization, and bioactivity. *Materials* **2020**, *13*, 394. [[CrossRef](#)] [[PubMed](#)]
36. Koutsopoulos, S. Synthesis and characterization of hydroxyapatite crystals: A review study on the analytical methods. *J. Biomed. Mater. Res.* **2002**, *62*, 600–612. [[CrossRef](#)] [[PubMed](#)]
37. Hong, L.; Wang, Y.L.; Jia, S.R.; Huang, Y.; Gao, C.; Wan, Y.Z. Hydroxyapatite/bacterial cellulose composites synthesized via a biomimetic route. *Mater. Lett.* **2006**, *60*, 1710–1713.
38. Catauro, M.; Tranquillo, E.; Poggetto, G.D.; Pasquali, M.; Dell’Era, A.; VecchioCipriotti, S. Influence of the heat treatment on the particles size and on the crystalline phase of TiO₂ synthesized by the sol-gel method. *Materials* **2018**, *11*, 2364. [[CrossRef](#)]
39. Klee, W.E.; Engel, G. IR spectra of the phosphate ions in various apatites. *J. Inorg. Nuclear Chem.* **1970**, *32*, 1837–1843. [[CrossRef](#)]
40. Kokubo, T.; Ito, S.; Huang, Z.T.; Hayashi, T.; Sakka, S.; Kitsugi, T.; Yamamuro, T. Ca, P-rich layer formed on high-strength bioactive glass-ceramic A-W. *J. Biomed. Mater. Res.* **1990**, *24*, 331–343. [[CrossRef](#)]
41. Kokubo, T.; Kushitani, H.; Sakka, S.; Kitsugi, T.; Yamamuro, T. Solutions able to reproduce in vivo surface-structure changes in bioactive glass-ceramic A-W3. *J. Biomed. Mater. Res.* **1990**, *25*, 721–734. [[CrossRef](#)]
42. Patlolla, A.; Knighten, B.; Tchounwou, P. Multi-walled carbon nanotubes induce cytotoxicity, genotoxicity and apoptosis in normal human dermal fibroblast cells. *Ethn Dis.* **2010**, *20* (Suppl. 1), S1–65–72.
43. Catauro, M.; Papale, F.; Bollino, F.; Piccolella, S.; Marciano, S.; Nocera, P.; Pacifico, S. Silica/quercetin sol-gel hybrids as antioxidant dental implant materials. *Sci. Technol. Adv. Mater.* **2015**, *16*, 035001. [[CrossRef](#)]
44. Yuan, X.; Zhang, X.; Sun, L.; Wei, Y.; Wei, X. Cellular Toxicity and Immunological Effects of Carbon-based Nanomaterials. *Part. Fibre Toxicol.* **2019**, *16*, 18. [[CrossRef](#)] [[PubMed](#)]
45. Ricciotti, L.; Roviello, G.; Tarallo, O.; Borbone, F.; Ferone, C.; Colangelo, F.; Catauro, M.; Cioffi, R. Synthesis and characterizations of melamine-based epoxy resins. *Int. J. Mol. Sci.* **2013**, *14*, 18200–18214. [[CrossRef](#)] [[PubMed](#)]
46. Pacifico, S.; Piccolella, S.; Papale, F.; Nocera, P.; Lettieri, A.; Catauro, M. A polyphenol complex from *Thymus vulgaris* L. plants cultivated in the Campania Region (Italy): New perspectives against neuroblastoma. *J. Funct. Foods* **2016**, *20*, 253–266. [[CrossRef](#)]
47. Gazzano, E.; Bracco, P.; Bistolfi, A.; Aldieri, E.; Ghigo, D.; Boffano, M.; Costa, L.; del Prever, E.B. Ultra high molecular weight polyethylene is cytotoxic and causes oxidative stress, even when modified. *Int. J. Immunopathol. Pharmacol.* **2011**, *24* (Suppl. 2), 61–67. [[CrossRef](#)] [[PubMed](#)]
48. Nguyen, K.; Garcia, A.; Sani, M.-A.; Diaz, D.; Dubey, V.; Clayton, D.; Poggetto, G.D.; Cornelius, F.; Payne, R.J.; Separovic, F.; et al. Interaction of N-terminal peptide analogues of the Na⁺,K⁺-ATPase with membranes. *BBA—Biomembranes* **2018**, *1860*, 1282–1291. [[CrossRef](#)]
49. Kiventer, J.; Lancellotti, I.; Catauro, M.; Poggetto, F.D.; Leonelli, C.; Illikainen, M. Alkali activation as new option for gold mine tailings inertization. *J. Clean. Prod.* **2018**, *187*, 76–84. [[CrossRef](#)]
50. Samal, S.; Kolinova, M.; Rahier, H.; Poggetto, G.D.; Blanco, I. Investigation of the internal structure of fiber reinforced geopolymer composite under mechanical impact: A micro computed tomography (μCT) study. *Appl. Sci.* **2019**, *9*, 516. [[CrossRef](#)]

51. Torrisi, L.; Visco, A.; Campo, N. Pulsed laser treatments of polyethylene films. *Nucl. Instrum. Meth. B* **2010**, *268*, 3117–3121. [[CrossRef](#)]
52. Cioffi, R.; Aronne, P.P.A.; Catauro, M.; Quattroni, G. Glass-Ceramics from fly ash with added Li₂O. *J. Eur. Ceram. Soc.* **1994**, *13*, 143–148. [[CrossRef](#)]



© 2020 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<http://creativecommons.org/licenses/by/4.0/>).